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Atomic-scale strain manipulation of a charge density wave

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A charge density wave (CDW) is one of the fundamental instabilities of the Fermi surface occurring in a wide range of quantum materials. In dimensions higher than one, where Fermi surface nesting can play only a limited role, the selection of the particular wavevector and geometry of an emerging CDW should in principle be susceptible to controllable manipulation. In this work, we implement a simple method for straining materials compatible with low-temperature scanning tunneling microscopy (STM/S), and use it to strain-engineer CDWs in 2H-NbSe₂. Our STM/S measurements, combined with theory, reveal how small strain-induced changes in the electronic band structure and phonon dispersion lead to dramatic changes in the CDW ordering wavevector and geometry. Our work unveils the microscopic mechanism of a CDW formation in this system, and can serve as a general tool compatible with a range of spectroscopic techniques to engineer electronic states in any material where local strain or lattice symmetry breaking plays a role.

Scanning tunneling microscopy | charge density waves | strain | NbSe₂

Significance

Charge density waves (CDWs) are simple periodic reorganizations of charge in a crystal, and yet they are still poorly understood and continue to bear surprises. External perturbations, such as strain or pressure, can in principle push a CDW phase into a different ordering geometry. However, engineering this type of quantum criticality has been experimentally challenging. Here, we implement a simple method for straining bulk materials. By applying it to 2H-NbSe₂, a prototypical CDW system studied for decades, we discover two dramatic strain-induced CDW phase transitions. Our atomic-scale spectroscopic imaging measurements, combined with theory, reveal the distinct roles of electrons and phonons in forming these emergent states, thus opening a window into the rich phenomenology of CDWs.


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(3Q) CDW ordering of ~3a₀ period (CDW-3a₀) below 33 K (10, 21, 28). In our strained samples of 2H-NbSe₂, in addition to detecting the well-known CDW-3a₀ in small patches (Fig. 1B), we reveal two additional types of charge ordering in other large regions of the sample—unidirectional “stripe” (1Q) ordering with 4a₀ period (CDW-4a₀) and a triangular (3Q) ordering with a 2a₀ period (CDW-2a₀) (Fig. 2 C and D). The wavevectors of all observed CDWs are found to be oriented along the Γ-M directions, based on the Fourier transforms of STM topographs where each CDW peak lies exactly along the atomic Bragg wavevector \( \mathbf{Q}_{\text{Bragg}} \) (Fig. 1 E–G). We have observed the same CDW wavevectors on multiple NbSe₂ single crystals attached to substrates with mismatched TECs (Methods). Interestingly, all of the CDW wavevectors measured are commensurate with the lattice, in contrast to the recently observed incommensurate 1Q CDW phase with an ~3.5a₀ period, which was found in accidentally formed nanometer-scale “ribbons,” and which could possibly be attributed to strain (21, 29). The magnitudes of the wavevectors identified in our experiments also do not change as a function of temperature \( T \) (SI Appendix, section I), which eliminates a dispersive quasiparticle interference (QPI) signal (10) as the cause of our observations.

The presence of multiple distinct CDWs in different regions of the same strained single crystal suggests that these phases may be associated with strain of locally varying magnitude and/or direction. Although in an ideal homogeneous sample attached to a substrate under elastic deformation, the strain is expected to remain laterally uniform as it is transmitted to the surface, this is unlikely to be the case in real materials that are inevitably inhomogeneous. In our NbSe₂ sample glued to a silica substrate by epoxy, inhomogeneous transmission of strain could arise due to the weak van der Waals interlayer bonding that makes the material prone to warping (4) or inhomogeneous glue distribution at the interface. To shed light on what type of strain, if any, might play a role in the formation of each observed CDW, it is necessary to quantify strain at the atomic length scales. We start with an STM topograph \( T(\mathbf{r}) \) to which we apply the transformation \( \mathbf{r} \rightarrow \mathbf{r} - \mathbf{u}(\mathbf{r}) \) [where \( \mathbf{u}(\mathbf{r}) \) is the total displacement field obtained from the Lawler–Fujita algorithm (30)], such that the resulting topograph \( T(\mathbf{r} - \mathbf{u}(\mathbf{r})) \) contains a perfect hexagonal lattice. We disentangle the experimental artifacts (piezo and thermal drift) from structural strain in \( \mathbf{u}(\mathbf{r}) \) by fitting and subtracting a polynomial background to create the strain field \( s(\mathbf{r}) \). The directional derivatives of \( s(\mathbf{r}) \) form a strain tensor \( s_{ij}(\mathbf{r}) \equiv \partial_s(\mathbf{r})/\partial r_i \) (where \( i, j = x, y \)), and their linear combinations provide information on the strain type and magnitude (31–33) (SI Appendix, section II). For example, we can extract biaxial (isotropic) strain as \( \sqrt{s_{xx} + s_{yy}}/2 \) (Fig. 2 C and D). Although this algorithm cannot provide us with the absolute value of the applied strain, it can extract the relative local strain variations between different regions within a single STM topograph. Applying this procedure to the occasionally encountered boundaries between the CDW-3a₀ and the newly observed CDW-2a₀ and CDW-4a₀ phases (Fig. 2 A and B), we find that regions hosting CDW-2a₀ and CDW-4a₀ are both under biaxial tensile strain (Fig. 2 C and D) with a prominent uniaxial strain component relative to the CDW-3a₀ phase (SI Appendix, section II). This is
Fig. 2. Local strain mapping. (A and B) STM topographs and (C and D) biaxial (isotropic) strain maps of the atomically smooth boundaries between regions hosting different CDW phases. The biaxial strain maps have been calculated from the derivatives of the strain fields as \( \varepsilon_{xx} + \varepsilon_{yy} \), using the procedure described in SI Appendix, section II. The algorithm assumes that strain is zero in the CDW-3a\(_0\) area, and calculates the relative strain with respect to it. Larger positive values represent tensile strain (stretching of the lattice). As can be seen, both CDW-2a\(_0\) and CDW-4a\(_0\) regions are characterized by tensile strain relative to the CDW-3a\(_0\) area. STM setup conditions were (A) \( I_{\text{set}} = 350 \) pA and \( V_{\text{sample}} = -70 \) mV; (B) \( I_{\text{set}} = 200 \) pA and \( V_{\text{sample}} = -100 \) mV.

Fig. 3. Electronic band-structure mapping using QPI imaging. FTs of \( \frac{dI}{dV} \) maps acquired at (A) \(-60\) mV, (B) \(0\) mV, and (C) \(60\) mV over the CDW-4a\(_0\) region of the sample. (B, Inset) Schematic of the Fermi surface under small tensile strain, which is expected to move the Fermi surface pockets around \( \Gamma \) further apart. Only \( Q_1 \) vector in \( E-G \) can be seen, while \( Q_2 \) is notably absent. (H) The dispersion of the QPI peaks as a function of energy along the \( \Gamma-M \) direction in the CDW-2a\(_0\) region. QPI peak positions in \( D \) and \( H \) are determined using Gaussian peaks fitting to a one-dimensional curve extracted along a line connecting the center of the FT and the atomic Bragg peak. QPI peaks and CDW peaks are denoted by the guides for the eye in panels A–C and E–G: \( Q_1 \) (green line), \( Q_2 \) (pink line), \( Q_{\text{CDW}} \) (orange circle), and \( Q_{\text{peak}} \) (brown square). The center of all FTs has been artificially suppressed to emphasize other features. All FTs have been sixfold symmetrized to enhance signal to noise, and cropped to the same \( 1.25 |Q| \) square size window. The region of the sample where the data in A–C were taken contains domains of CDW-4a\(_0\) along only two lattice directions (SI Appendix, Fig. S2A). As CDW-4a\(_0\) is intrinsically a unidirectional order, the sixfold symmetry of the \( Q_{\text{peak}} \) peak in \( A-C \) is an artifact of the symmetrization process. STM setup conditions: (A–C) \( I_{\text{set}} = 320 \) pA, \( V_{\text{sample}} = -60 \) mV, and \( V_{\text{background}} = 10 \) mV (zero-to-peak); (E) \( I_{\text{set}} = 200 \) pA, \( V_{\text{sample}} = -39 \) mV, and \( V_{\text{background}} = 1 \) mV; (F) \( I_{\text{set}} = 20 \) pA, \( V_{\text{sample}} = 5 \) mV, and \( V_{\text{background}} = 1.5 \) mV; (G) \( I_{\text{set}} = 300 \) pA, \( V_{\text{sample}} = 50 \) mV, and \( V_{\text{background}} = 10 \) mV.

direct proof that in-plane tensile strain plays an important role in driving the observed charge ordering transitions.

To gain insight into the effects of strain on local electronic band structure in each region of the sample, we use QPI imaging, a method that applies 2D Fourier transforms (FTs) to the STM \( \frac{dI}{dV} \) maps to extract the electronic band dispersion. First, we focus on a large region of the sample hosting exclusively CDW-4a\(_0\), in which the FTs of the \( \frac{dI}{dV} \) maps show a circular QPI morphology (Fig. 3 A–C) with the strongest intensity along the \( \Gamma-M \) direction. Higher momentum-space resolution of our data compared with previous experiments on NbSe\(_2\) hosting a CDW-3a\(_0\) (10) allows us to distinguish two different QPI peaks \( Q_1 \) and \( Q_2 \) (Fig. 3B), which arise from backscattering within the two Fermi surface pockets concentric around \( \Gamma \) (Fig. 3B, Inset and SI Appendix, section III). By measuring the positions of these peaks as a function of energy, we can map the two bands crossing the Fermi level along the \( \Gamma-M \) direction (Fig. 3D). Interestingly, the electronic band structure is only slightly different compared with that of the well-characterized unstrained material (10) (SI Appendix, section IV), despite the dramatic changes in both the observed CDW wavelength and its geometry.

In the CDW-2a\(_0\) region, we observe only the \( Q_1 \) vector, while \( Q_2 \) is notably absent in our measurable momentum range, in contrast to the CDW-4a\(_0\) area (Fig. 3 E–G). This suggests a more prominent change in the band structure. Our strain measurements in Fig. 2 reveal that this region of the sample is under tensile strain, which would lead to a larger momentum-space separation of the pockets around \( \Gamma \) (Fig. 3F, Inset), owing to
the concomitant increase in the interlayer tunneling (as the interlayer orbital overlaps increase). Our OPI measurements however have been unable to detect any scattering vectors larger than $\mathbf{Q}_{\text{Bragg}}/2$ in either CDW-2$_{a_0}$ or CDW-4$_{a_0}$ regions at any energy (SI Appendix, section V), and we therefore cannot directly observe the shift of $Q_z$ to higher momenta. A possible explanation for the lack of signal at higher momenta may be canted of the orbital texture toward more in-plane orientations (34), making them less likely to be detected by the STM tip. Nevertheless, our measurements reveal that a larger distortion to the Fermi surface accompanies the formation of a CDW-2$_{a_0}$.

**Discussion**

Having quantified the changes in the structural and electronic properties of regions hosting CDW-2$_{a_0}$ and CDW-4$_{a_0}$, we turn to the fundamental question of what drives and stabilizes a particular CDW wavevector and geometry in this quasi-2D system. Taking into account the exactly commensurate nature of all observed CDWs, Fermi surface nesting is even more unlikely to play a role for the observed CDW phases. To provide further insight, we construct a simple model that captures the strain effects on both the electronic structure and phonon dispersion.

We start with a tight-binding fit to the angle-resolved photoemission spectroscopy (ARPES) data (26, 35), including the in-plane strain by modifying the hopping integrals, and employ the Random Phase Approximation to calculate the resulting full electronic susceptibility $D_2(q)$ (Methods and SI Appendix, section VI). We separately introduce the effect of the uniaxial strain on the phonons by shifting their bare energies differently in lattice-equivalent directions (29). Within this model’s description, the CDW ordering vector can be identified as the first wavevector for which the calculated susceptibility $D_2(q)$ exceeds the bare phonon energy $\Omega(q)$ identified in resonant inelastic X-ray scattering experiments (20, 36).

In our model, we consider the effects of both uniaxial and biaxial in-plane strain, each modeled by a relative change in the nearest-neighbor overlap integrals $\sigma$ associated with the uniaxial strain and $\sigma_i$ associated with the biaxial strain (for more details, see Methods and SI Appendix, section VI). For simplicity, we explore the effects of the two types of strain separately. We find that biaxial strain by itself has very little effect on the shape of $D_2(q)$, while the uniaxial strain can lead to a significant change in $D_2(q)$ and induce different types of CDW ordering (Fig. 4).

Specifically, we find that $\sigma = 0.1$ (stretching along $\Gamma$-M and compressing along the perpendicular $\Gamma$-K direction) stabilizes the CDW-4$_{a_0}$ order, with a peak in $D_2(q)$ forming near $0.25|\mathbf{Q}_{\text{Bragg}}|$ and $0.28|\mathbf{Q}_{\text{Bragg}}|$ momentum transfer wavevector (Fig. 4). The predicted CDW geometry is 3Q, but inclusion of anisotropy in the phonon energies of around 1.8%, the same order of magnitude as the strain, is enough to yield the experimentally observed 1Q state. Similarly, we find that $\sigma = -0.3$ (stretching along $\Gamma$-K and compressing along the perpendicular $\Gamma$-M direction) leads to a CDW with a peak in $D_2(q)$ forming near $0.4|\mathbf{Q}_{\text{Bragg}}|$ (Fig. 4). In this case, the energetic payoff of locking into the nearest commensurate structure (37), which is not included in the present model, would be expected to increase the CDW wavevector to the observed CDW-2$_{a_0}$ period. While it is difficult to obtain the exact relationship between $\sigma/\sigma_i$ and the magnitude of real-space lattice distortion, the generic dependence of the orbital overlap on interatomic distance found in, for example ref. (38), suggests that changes in the overlap integrals are expected to be approximately five times the relative strain as defined in the experimental analysis. Using this rough estimate, we calculate the magnitude and direction of strain used in our model to achieve different CDWs, which leads to a reasonable agreement with the relative strain values observed in the experiment (SI Appendix, section VI). Moreover, the electronic band dispersion used to calculate $D_2(q)$ in the presence of these strain levels presents a good match to the experimentally measured electronic dispersion obtained from the OPI data in Fig. 3. Remarkably, the calculations indicate that both 1Q and 3Q phases of CDW-2$_{a_0}$ may be stabilized, which can in fact be observed in STM data acquired at higher bias (SI Appendix, section VIII). Despite its simplicity, our model is able to reproduce the wavevectors and geometries of all observed CDWs, and points to the dominant physical mechanism behind the CDW formation. CDW order is sensitive to two effects of strain—softening of phonon energies and modification of electron-hopping parameters—each playing a distinct role in the formation of the resulting CDW phase. The main effect of the changes in the phonon dispersion by strain is the favoring of one type of geometry (stripe 1Q) over another (triangular 3Q). The effect of the electronic modification, on the other hand, is to alter the CDW wavevector, and even relatively small strain can have a significant effect. Exploiting these trends, we should in principle be able to strain-engineer desired charge-ordering structures in this and other materials by considering the shift in the peak in the electronic susceptibility.

Our simple platform for exerting strain on bulk single crystals presented here can be combined with a variety of characterization techniques. A single CDW domain can be found over microscopically large regions of the sample covering hundreds of nanometers (SI Appendix, section VIII), so in addition to nanoscale methods, micro-ARPES or micro-Raman spectroscopy could also be used to study these phases. Moreover, this strain
technique can be applied to a range of other materials. For example, 1T-TiSe₂ could be induced to change superconductivity (39) or novel CDW wavevector and geometries in analogy to what we observe in 2H-NbSe₂. Similarly, Fe-based superconductors could be strained, potentially using substrates with a TEC along a preferred direction (3), to create a rich playground to study the interplay of nematic order and superconductivity (40) within a single material using SI-STM.

Methods

Single crystals of 2H-NbSe₂ were grown using vapor transport growth technique with iodine (I₂) as the transport agent, and exhibit superconducting transition temperature Tc ~ 7 K based on the onset of diamagnetic signal due to the Meissner effect in magnetization measurements (SI Appendix, section IX). Superconducting transition temperature remained approximately the same with Tc ~ 7 K after the samples were strained and remeasured. Typical size of the single crystals used was ~2 mm × 2 mm, with ~0.1-mm thickness before cleaving and ~0.01-mm to ~0.1-mm thickness postcleaving. Instead of attaching the 2H-NbSe₂ crystals directly to a metallic holder with TEC comparable to that of NbSe₂, as typically used in most STM experiments, we use conducting epoxy (EPO-TEK H20E) to glue the bottom of NbSe₂ to silica (SiO₂) material with a vastly different TEC (Fig. 1A). Then, the sample structure is attached to the STM sample holder and cooled down to ~4.5 K (more information in SI Appendix, section IX). Based on the difference between TECs of NbSe₂ and silica, NbSe₂ is expected to stretch isotropically in-plane by ~0.15%. As we demonstrate from STM topographs, the actual induced strain in the sample surface can be spatially inhomogeneous; to create a clean surface necessary for STM measurements, the samples were cleaved in ultra-high vacuum (UHV), and inserted into the STM head within minutes. We studied four different NbSe₂ crystals glued on silica (five different surfaces as one sample was reclaved for the second approach). For each of these five, we approached the tip on several different points on the sample, which are typically tens of micrometers away from one another, and searched for different types of CDWs. We observed: all three CDWs on two surfaces, just CDW-2a and CDW-4a on two other surfaces, and just CDW-3a on one surface.

STM data were acquired using a Uniosoku USM1300 STM at the base temperature of ~4.5 K. All spectroscopic measurements have been taken using a standard lock-in technique at 915-Hz frequency and varying bias excitation as detailed in the figure legends. The STM tips used were home-made, chemically etched W tips annealed to bright-orange color in UHV. Tip quality has been evaluated on the surface of single-crystal Cu(111) before performing the measurements presented in this paper. The Cu(111) surface was cleaned by repeated cycles of heating and argon sputtering in UHV before it was inserted into the STM head.

To construct a model which captures experimental observations, we employ a tight-binding fit to the ARPES data for the two bands crossing the Fermi level (described in detail in refs. 26 and 35). The model assumes the two bands to be bonding and antibonding combinations of the two Nb dₓzy-orbitals. We include both biaxial and uniaxial in-plane strain by modifying the hopping integrals based on the assumption that overlap integrals are linearly dependent on displacement, with an equal prefactor for all overlap. In modeling uniaxial strain, we assume that a tensile strain in one direction leads to a compressive strain in the perpendicular in-plane direction, conserving the volume of the unit cell. Then, we employ the Random Phase Approximation to calculate the phonon softening as seen in resonant inelastic X-ray scattering (20, 36). The CDW wavevector is identified as the first wavevector to soften to zero. By including nonlinear terms in a Landau free-energy expression we are able to reveal whether the CDW geometry consists of stripes (10) or triangles (30) (see SI Appendix, section VI for more details).

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