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Thermodynamically Consistent Methodology for Estimation of Diffusivities of Molecules in Microporous Materials

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Supporting Information

ABSTRACT: The Maxwell–Stefan (M–S) formulation, that is grounded in the theory of irreversible thermodynamics, is widely used for describing mixture diffusion in microporous crystalline materials such as zeolites and metal–organic frameworks (MOFs). Binary mixture diffusion is characterized by a set of three M–S diffusivities: D_{12}, D_{12}, and D_{12}. The M–S diffusivities D_{11} and D_{22} characterize interactions of guest molecules with pore walls. The exchange coefficient D_{12} characterizes correlation effects that result in slowing-down of the more mobile species due to correlated molecular jumps with tardier partners. The primary objective of this article is to develop a methodology for estimating D_{11}, D_{22}, and D_{12} using input data for the constituent unary systems. The dependence of the unary diffusivities D_{11} and D_{22} on the pore occupancy, \theta, is quantified using the quasi-chemical theory that accounts for repulsive, or attractive, forces experienced by a guest molecule with the nearest neighbors. For binary mixtures, the same occupancy dependence of D_{11} and D_{22} is assumed to hold; in this case, the occupancy, \theta, is calculated using the ideal adsorbed solution theory. The exchange coefficient D_{12} is estimated from the data on unary self-diffusivities. The developed estimation methodology is validated using a large data set of M–S diffusivities determined from molecular dynamics simulations for a wide variety of binary mixtures (H_{2}/CO_{2}, Ne/CO_{2}, CH_{4}/CO_{2}, CO_{2}/N_{2}, H_{2}/CH_{4}, H_{2}/Ar, CH_{4}/Ar, Ne/Ar, CH_{4}/C_{2}H_{4}, CH_{4}/C_{3}H_{8}, and C_{2}H_{6}/C_{3}H_{8}) in zeolites (MFI, BEA, ISV, FAU, NaY, NaX, LTA, CHA, and DDR) and MOFs (IRMOF-1, CuBTC, and MgMOF-74).

1. INTRODUCTION

Many separation and reaction processes use microporous crystalline materials such as zeolites (alumino-silicates), metal–organic frameworks (MOFs), and zeolitic imidazolate frameworks (ZIFs) as perm-selective membrane layers, adsorbents, or catalysts.\textsuperscript{1–9} Separation of nitrogen/oxygen, nitrogen/methane, and propene/propane mixtures in adsorbents packed with LTA-4A, CHA, or ZIF-8 are essentially driven by differences in the pore diffusivities of the guest constituents.\textsuperscript{9–16} The conversion and selectivity of several heterogeneous catalytic reactions are influenced by intra-particle transport of reactants and products.\textsuperscript{17–19} Pore diffusion characteristics have a significant influence on membrane separation selectivities.\textsuperscript{4,10–19}

Mixture diffusion in microporous materials is characterized by the fact that the mobility of any guest constituent is influenced by its partner species;\textsuperscript{20,26} the proper modeling of such influences is essential for process development and design.\textsuperscript{5,15} It is a common practice to model n-component mixture diffusion by adopting the Maxwell–Stefan (M–S) formulation,\textsuperscript{16,7,27,28} that has its foundations in the theory of nonequilibrium thermodynamics. The dependence of the intracrystalline molar fluxes, \( N_j \), on the chemical potential gradients is written in the following form

\[
-\frac{\rho}{RT} \frac{dn_j}{dz} = \sum_{i=1}^{n} x_i N_j - x_j N_i \frac{D_{ij}}{D_{ji}} + \frac{N_j}{D_{ji}} \quad i = 1,2,..n
\]

In eq 1, \( R \) is the gas constant, \( \rho \) represents the material density, and the component loadings, and \( q_i \) are defined in terms of moles per kg of framework material. The \( x_i \) in eq 1 is the mole fractions of the adsorbed phase components

\[
x_i = \frac{q_i}{\rho} \quad q_i = q_1 + q_2 + ... q_n \quad i = 1,2,..n
\]

Two distinct sets of M–S diffusivities are defined by eq 1, that is phenomenological in nature.\textsuperscript{29} The \( D_j \) characterize interactions between species \( i \) with the pore walls. As established in earlier works,\textsuperscript{23,28} the important advantage of the M–S formulation is that the \( D_j \) can be identified with the corresponding unary diffusivities, provided the diffusivity data are compared at the same adsorption potential, \( \pi A/RT \), where \( A \) represents the surface area per kg of framework material, and \( \pi \) is the spreading pressure, defined by the Gibbs adsorption equation.\textsuperscript{3,30,31}

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The ideal adsorbed solution theory (IAST) of Myers and Prausnitz enables the calculation of the adsorption potential, $\pi A/RT$; details are provided in the Supporting Information.

As illustration, Figure 1 presents plots of the transport coefficients, $\rho D/D_i$, of CO$_2$, CH$_4$, N$_2$, and H$_2$ determined for unary and equimolar binary CO$_2$/H$_2$, CO$_2$/CH$_4$, CO$_2$/N$_2$, CH$_4$/Ar, CH$_4$/N$_2$, CH$_4$/H$_2$, and N$_2$/H$_2$ mixture permeation across SAPO-34 membrane at 295 K. The data are plotted as a function of the adsorption potential, $\pi A/RT$, calculated for conditions prevailing at the upstream face of the membrane; the calculation details are provided in an earlier work.

\[
A \, dx = \sum_{i=1}^{n} \eta_i \, d\mu_i \tag{3}
\]

The ideal adsorbed solution theory (IAST) of Myers and Prausnitz enables the calculation of the adsorption potential, $\pi A/RT$; details are provided in the Supporting Information.

As illustration, Figure 1 presents plots of the transport coefficients, $\rho D/D_i$, of CO$_2$, CH$_4$, N$_2$, and H$_2$ determined for unary and equimolar binary (CO$_2$/H$_2$, CO$_2$/CH$_4$, CO$_2$/N$_2$, CH$_4$/Ar, CH$_4$/N$_2$, CH$_4$/H$_2$, and N$_2$/H$_2$) mixture permeation across SAPO-34 membrane of thickness $\delta$. SAPO-34 has the same structural topology as CHA zeolite, consisting of cages of volume 316 Å$^3$, separated by 8-ring windows of 3.8 × 4.2 Å size. Compared at the same value of $\pi A/RT$, the magnitude of $\rho D/D_i$ for a binary mixture is comparable to that for the corresponding pure component. Also noteworthy, from the data in Figure 1 is that the dependence of $\rho D/D_i$ on $\pi A/RT$ is distinctly different for each guest molecule.

The exchange coefficients, $D_{ij}$, defined by the first right member of eq 1 reflect how the facility for transport of species $i$ correlates with that of species $j$. The Onsager reciprocal relations impose the symmetry constraint

\[
D_{ij} = D_{ji} \tag{4}
\]

The magnitude of $D_i$ relative to that of $D_{ij}$ determines the extent to which the flux of species $i$ is influenced by the chemical potential gradient of species $j$. The larger the degree of correlations, $D_i/D_{ij}$, the stronger is the influence of diffusional “coupling”. Generally speaking, the more strongly adsorbed tardier partner species will have the effect of slowing down the less strongly adsorbed more mobile partner in the mixture.

For estimation of the exchange coefficient, $D_{ij}$, the following interpolation formula has been suggested in the literature:

\[
D_i = (D_{ii})^{y/(x+y)}(D_{ij})^{y/(x+y)} \tag{5}
\]

where the $D_{ii}$ and $D_{ij}$ represent the self-exchange coefficients, that are accessible from molecular dynamics (MD) simulations of self-diffusivities for the constituent unary systems, as will be discussed in a subsequent section. Equation 5 is essentially an adaptation of the interpolation formula for estimation of the $\mathbf{M}$–$\mathbf{S}$ diffusivity for binary fluid mixtures.

Specifically, for a binary mixture, that is $n = 2$, the $\mathbf{M}$–$\mathbf{S}$ eq 1 can be rewritten to evaluate the fluxes $N_i$ explicitly by defining a $2 \times 2$ dimensional square matrix $[\Lambda]$

\[
N_i = -\rho \sum_{J=1}^{2} \Lambda_{ij} \frac{q_i}{RT} \, dz; \quad i = 1, 2 \tag{6}
\]

The elements of $[\Lambda]$ are directly accessible from MD simulations by monitoring the individual molecular displacements.
In 7, $n_i$ and $n_j$ represent the number of molecules of species $i$ and $j$, respectively, and $r_l(t)$ is the position of molecule $l$ of species $i$ at any time $t$.

Combining eq 1 with eq 6, the following explicit expression for calculation of the elements of the $2 \times 2$ dimensional square matrix $[\Lambda]$ can be derived

$$[\Lambda] = \frac{1}{1 + \frac{x_{D_1}D_2}{D_{12}} + \frac{x_{D_1}D_2}{D_{12}}} \begin{bmatrix} D_i \left(1 + \frac{x_{D_1}D_2}{D_{12}}\right) & x_{D_1}D_2 \frac{x_{D_1}D_2}{D_{12}} \\ x_{D_1}D_2 \frac{x_{D_1}D_2}{D_{12}} & D_i \left(1 + \frac{x_{D_1}D_2}{D_{12}}\right) \end{bmatrix} \right) \right)^2$$

(8)

The primary objective of this article is to seek validation of the predictive capability of the Maxwell–Stefan formulation by comparing each of the four elements, $\Lambda_{ij}$, determined from MD simulations using eq 7, with the estimations using eqs 5 and 8; the required data inputs for $D_1$, $D_2$, $D_{11}$, and $D_{22}$ are determined from MD simulations for the constituent pure components. To meet the stated objective, use is made of the MD simulation data base compiled in an earlier work for a variety of binary mixtures (H$_2$/CO$_2$, Ne/CO$_2$, CH$_4$/CO$_2$, CO$_2$/N$_2$, H$_2$/CH$_4$, H$_2$/Ar, CH$_4$/Ar, Ne/Ar, CH$_4$/C$_2$H$_6$, CH$_4$/C$_3$H$_8$, and C$_2$H$_6$/C$_3$H$_8$) in different host materials. The host materials were chosen to represent a variety of pore sizes, topologies, and connectivities: one-dimensional (1D) channels (e.g., AFI, MgMOF-74), intersecting channels (MFI, BEA, ISV), cages separated by narrow windows (LTA, CHA, DDR), cavities with large windows (FAU, NaY, NaX, IRMOF-1, CuBTC). The Supporting Information provides structural details for all of the host materials considered in this article.

2. RESULTS AND DISCUSSIONS

2.1. Occupancy Dependence of Unary Diffusivities.

For any guest molecule, the loading dependence of $D_i$ is strongly influenced also by the pore topology and connectivity and molecule–molecule interactions. As an illustration, Figure 2a,b presents data on $D_i$ for the guest species CH$_4$ in a variety of host structures, determined from MD simulations of molecular displacements using the following formula in each of the coordinate direction

$$D_i = \frac{1}{2} \lim_{\Delta t \to \infty} \frac{1}{n_i \Delta t} \left\{ \sum_{l=1}^{n_i} (r_{ij}(t + \Delta t) - r_{ij}(t)) \right\}^2$$

(9)

The M–S diffusivity, $D_i$, displays a wide variety of dependencies on the fractional occupancy, $\theta$, that serves as a convenient and practical proxy for the adsorption potential

$$\theta \equiv 1 - \exp \left(-\frac{\pi A}{q_{sat} RT} \right)$$

(10)

where $q_{sat}$ is the saturation capacity. The calculations of the adsorption potential in eq 10 use dual-site Langmuir–
Freundlich fits of the unary isotherms that are determined from configurational-bias Monte Carlo (CBMC) simulations 44–49 for each guest/host combination; details are provided in the Supporting Information.

For CH4/BEA and CH4/NaX, the D1 appears to decrease almost linearly with occupancy 0 till pore saturation conditions, 0 = 1, are reached. An appropriate model to describe this occupancy dependence is

\[ D_1 = D_1(0)(1 - \theta) \]

(11)

where \( D_1(0) \) is the M–S diffusivity at “zero-loading”. Equation 11 is essentially based on a hopping model in which one molecule at any time can jump from one sorption site to an adjacent one, provided it is not already occupied. 25,28,50

Using a simple two-dimensional square lattice model, the M–S diffusivity in the limit of vanishingly small occupancies, \( D_1(0) = \frac{1}{4} \nu(0) \lambda^2 \), where \( \lambda = 4 \) is the coordination number of the 2D array of lattice sites, \( \lambda \) is the jump distance on the square lattice, and \( \nu(0) \) is the jump frequency at vanishingly small occupancy. 50

More generally, molecule–molecule interactions serve to influence the jump frequencies by a factor that depends on the energy of interaction, \( w \). For repulsive interactions, \( w > 0 \), whereas for attractive interactions, \( w < 0 \). Using the quasi-chemical approach of Reed and Ehrlich 51 to quantify such interactions, the following expression is obtained for the occupancy dependence of the M–S diffusivities 50,52,53

\[ D_1 = D_1(0) \left( \frac{1 + \beta}{2(1 - \theta)} \right)^{\frac{1}{z}} \left( 1 + \frac{(\beta - 1 + 2\theta)\phi}{2(1 - \theta)} \right)^{\frac{z - 1}{z}} \]

(12)

In 12 the following dimensionless parameters are defined

\[ \beta = \sqrt{1 - 4\theta (1 - \theta)}(1 - 1/\phi); \quad \phi = \exp(w/RT) \]

(13)

In the limiting case of negligible molecule–molecule interactions, \( w = 0 \), \( \phi = 1 \), \( \beta = 1 \), eqs 12 and 13 degenerate to yield eq 11. The continuous solid lines in Figure 2a,b are fits of the MD simulated \( D_1 \) by fitting the sets of parameters: \( D_1(0) \), and \( \phi = \phi_0 \exp(-aw) \). For all of the guest/host combinations, eqs 12 and 13 provide good descriptions of the occupancy dependencies; see Figures S38–S108 of the Supporting Information.

Applying eq 1 to a binary mixture consisting of tagged and untagged species \( i \), that are otherwise identical, 1,50,54 we can derive the following relation between the self-diffusivity, \( D_{1,\text{self}} \) and the M–S diffusivity, \( D_1 \)

\[ \frac{1}{D_{1,\text{self}}} = \frac{1}{D_1} + \frac{1}{D_i} = \frac{1}{D_1} \left( 1 + \frac{D_i}{D_{1i}} \right) \]

(14)

The self-diffusivities, \( D_{1,\text{self}} \) may be computed from MD simulations by analyzing the mean square displacement of each species, \( i \) for each coordinate direction

\[ D_{1,\text{self}} = \frac{1}{2n_1 \Delta t} \lim_{\Delta t \to \infty} \frac{1}{\Delta t} \left( \sum_{t=1}^{n_1} (r_{1i}(t + \Delta t) - r_{1i}(t))^2 \right) \]

(15)

By combination of eqs 9, 14, and 15, we can determine the degrees of correlations due to self-exchange, \( D_i/D_1 \). Figure 2c,d presents MD simulated data on \( D_{1,\text{self}} \) and \( D_1/D_i \) for CH4 in different host materials: MgMOF-74 (1D channels of 11 Å), BEA (intersecting channels of 6.5 Å), ISV (intersecting channels of 6.5 Å), NaX (790 Å2 cages separated by 7.4 Å windows), MFI (intersecting channels of 5.5 Å), and LTA (743 Å2 cages separated by 4.2 Å windows). It is also to be noted that the size of the 1D channels of MgMOF-74 are large enough to preclude single-file diffusion of guest molecules. The degree of correlations is the lowest for the LTA zeolite because the guest molecules jump one-at-a-time across the narrow 4.2 Å windows, 6,55,56 the same characteristics are valid for other cage-window structures with narrow windows, such as CHA, DDR, and ZIF-8. 24,57–59 The variation of \( D_i/D_1 \) with occupancy is practically linear, 25,28 and the solid lines in Figure 2d are the linear fits

\[ D_i/D_1 = b \theta \]

(16)

Equation 16 provides a good description of the occupancy dependence of the degrees of correlations due to self-exchange for all guest/host combinations; see Figures S38–S108 of the Supporting Information.

2.2. Occupancy Dependence of \( \Lambda_{ij} \) for Binary Mixture Diffusion. Having established and quantified the occupancy dependence of \( D_i \) and \( D_i/D_1 \) for each guest/host combination, we are in a position to compare the estimations of \( \Lambda_{ij} \) for binary mixtures using eqs 5 and 8 with the corresponding MD simulated values by monitoring molecular
displacements and use of eq 7; Figures S38–S108 provide detailed comparisons for each mixture/host combination that was investigated. Figure 3 provides an illustration of the estimation procedure for CH4(1)/C3H8(2) mixture diffusion in NaX zeolite. Figure 3a show the Reed–Ehrlich model fits for the unary diffusivities \( \lambda_{11} \) and \( \lambda_{22} \) for CH4 and C3H8 in NaX. The linear fits for the degrees of self-exchange \( \lambda_{11}/\lambda_{11} \) and \( \lambda_{22}/\lambda_{22} \) are shown in Figure 3b. In Figure 3c, the MD simulation data for \( \Lambda_{11}, \Lambda_{12} = \Lambda_{21}, \) and \( \Lambda_{22} \) for equimolar \( (q_1 = q_2) \) binary CH4(1)/C3H8(2) mixtures are compared with the estimations (shown by continuous solid lines) using eqs 5 and 8. The Maxwell–Stefan formulation provides very good estimates of dependence of each \( \Lambda_{ij} \) on the occupancy \( \theta_i \), calculated using eq 10, wherein the saturation capacity for the mixture is determined from

\[
\frac{1}{q_{1,\text{sat}}} = \frac{x_1}{q_{1,\text{sat}}} + \frac{x_2}{q_{2,\text{sat}}}
\]  

(17)

where \( q_{1,\text{sat}} \) and \( q_{2,\text{sat}} \) are the saturation capacities of components 1 and 2, respectively. Equation 17 can be derived from the IAST, as detailed in the Supporting Information.

Similar good estimates of the M–S model are established in Figure 4 for six other equimolar \( (q_1 = q_2) \) binary mixtures: CO2/H2 in MFI, CH4/C3H8 in BEA, CH4/C2H6 in NaY, CH4/CO2 in IRMOF-1, CO2/CH4 in MgMOF-74, and Ne/Ar in CuBTC.

A different test of the predictive capability of M–S formulation is to consider diffusion in binary mixtures for which the total loading \( q_1 + q_2 \) is held constant, and the mole fraction of component 1 in the adsorbed mixture, \( x_1 \), is varied from 0 to 1.\(^{60}\) One of the earliest investigations of this type were reported by Snurr and Kärger\(^{61}\) for CH4/CF4 diffusion in MFI zeolite at a total loading of 12 molecules uc\(^{-1}\).

Figure 5 compares the MD simulation data for \( \Lambda_{ij} \) for binary Ne(1)/Ar(2) mixtures of varying composition \( x_1 \) in MFI, LTA, CHA, and DDR zeolites. In all four cases, eqs 5,
and 8 provide good predictions of the variation of $\Lambda_{ij}$ with composition. It is also to be noted that the off-diagonal elements $\Lambda_{12}$ for LTA, CHA, and DDR zeolites are significantly lower, by about an order of magnitude, than the diagonal elements, $\Lambda_{11}$ and $\Lambda_{22}$. For cage-type zeolites such as LTA, CHA, DDR, ERI with 8-ring windows in the 3.3−4.5 Å size range, the degree of correlations $\mathcal{D}_i/\mathcal{D}_{ii}$ are negligibly small because the guest molecules jump one-at-a-time across the narrow windows.20,26−28,55,62 Further evidence of the good predictive capability of the M−S formulation is provided in Figures S38−S108.

2.3. Preferential Perching of CO2 in Window Regions of Cage-Type Zeolites. For separation of CO2 from gaseous mixtures containing CH4, H2, N2, Ar, or Ne, cage-type zeolites such as DDR, CHA, LTA, and ERI are of practical interest.8,24,30,31,46,47,55,56 These materials consist of cages separated by narrow windows, in the 3.3−4.5 Å range. CBMC simulations59 show that the window regions of cage-type zeolites have a significantly higher proportion of CO2 than within the cages. For all four zeolites, CO2 has the highest probability, about 30−40%, of locating at the window regions.59 The preferential perching of CO2 in the window regions, evidenced by the computational snapshot for CHA (see Figure 6a), has the effect of hindering the intercage hopping of partner molecules.59

Figure 6b−d compare the MD simulation data for $\Lambda_{11}$ and $\Lambda_{22}$ for equimolar ($q_1 = q_2$) binary CO2(1)/Ne(2) mixtures in LTA, CHA, and DDR zeolite, with the estimations using eq 8, assuming that the degrees of correlations are negligible, that is $D_i/D_{ii} \rightarrow 0$. For all three zeolites, the MD simulation data for $\Lambda_{22}$ are significantly lower than the predictions using eq 8; the M−S formulation does not cater for hindering effects caused due to segregated mixture adsorption. Experimental evidence of the importance of hindering effects is provided in published works on CO2/CH4 and CO2/N2 mixture permeation across the DDR membrane.38,63,64 Analogous hindering effects are also evidenced for CO2/CH4 mixture permeation across ZIF-8 membranes.24

Preferential location of branched alkanes and aromatics at the intersections of MFI zeolite often cause intersection-blocking and loss of connectivity;55,65,66 this leads to failure of the predictions of the M−S model.67

2.4. Molecular Clustering Due to Hydrogen Bonding. For water/methanol and water/ethanol mixture diffusion in microporous materials, molecular simulations62,68−71 demonstrate the occurrence of molecular clustering due to hydrogen bonding. As a consequence of cluster formation, the diffusivities of either guest molecule in the mixture is significantly lower than the corresponding unary diffusivities. As illustration of mutual-slowing down effects, Figure 7 presents MD data on the self-diffusivities, $D_{self}$ in water/methanol mixtures in CHA, DDR, and LTA zeolites, plotted as a function of the mole fraction of water in the adsorbed phase, $x_1$. Each of the diffusivities is lowered due to the presence of its partner species. Experimental evidence of mutual-slowing down effects are available for water/alcohol permeation across CHA,72,73 H-SOD,74 and DDR71,75 membranes. Further research is necessary to generalize the
M–S formulation in a manner that explicitly allows for cluster formation, by defining a cluster to be a pseudospecies in the mixture.

3. CONCLUSIONS

The capability of eqs 5 and 8 for the estimation of the elements of the square matrix of M–S diffusivities, $\Lambda_{ij}$ characterizing mixture diffusion, using input based on unary systems is tested using a large database obtained from MD simulations for a wide variety of guest/host combinations. The key to the estimation methodology is that the estimates are based on comparing the mixture diffusion data with those of the constituent unaries at the same fractional occupancy, $\theta$, that is calculated on the basis of the adsorption potential using eq 10. For the majority of binary mixtures investigated, 70 in total, summarized in Figures S38–S108, the MD-simulated $\Lambda_{ij}$ data are in good agreement with the estimations using eq 8 and assuming that the degrees of correlation are negligible, that is $\mathcal{D}_i/\mathcal{D}_i \to 0$.

Figure 6. (a) Computational snapshot showing the preferential perching of CO$_2$ at the window regions of CHA zeolite.$^{59}$ (b–d) MD simulation data for $\Lambda_{11}$ and $\Lambda_{22}$ for equimolar ($q_1 = q_2$) binary CO$_2$(1)/Ne(2) mixtures at 300 K (b) LTA all-silica zeolite, (c) CHA all-silica zeolite, and (d) DDR zeolite, compared with the estimations (shown by continuous solid lines) using eq 8 and assuming that the degrees of correlations are negligible, that is $\mathcal{D}_i/\mathcal{D}_i \to 0$.

Figure 7. MD simulations of self-diffusivities, $D_{\text{self}}$, of water(1)/methanol(2) mixtures of varying composition in (a) CHA, (b) DDR, and (c) LTA zeolites, plotted as a function of the mole fraction of water in the adsorbed phase, $x_1$. In the MD simulations, the total loading, $\Theta_t$, expressed as molecules uc$^{-1}$, is held constant; the values $\Theta_t$ are specified. The MD data are culled from our previous publications.$^{69}$–$^{71}$
to capture molecular clustering effects in water/alcohol systems that are engendered due to hydrogen bonding.62

ASSOCIATED CONTENT

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsomega.9b01873.

Structural details for zeolites and MOFs considered and analyzed in this article, dual-Langmuir–Freundlich data fits of unary isotherms for each guest/host combination, IAST calculation procedure for the spreading pressure and the fractional occupancy, θ, and MD simulation data (a total of 70 data sets) for Λ_{0i} along with comparisons with estimates using eqs S, and 8 (PDF)

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NOMENCLATURE

Latin alphabet

A surface area per kg of framework, m² kg⁻¹

D_i Maxwell–Stefan diffusivity for molecule---wall interaction, m² s⁻¹

D_i(0) M--S diffusivity at zero-loading, m² s⁻¹

D_0 M--S exchange coefficient, m² s⁻¹

D_i M--S self-exchange coefficient, m² s⁻¹

D_0 i self-diffusivity of species i, m² s⁻¹

n number of species in the mixture, dimensionless

n_i number of molecules of species i in simulation box, dimensionless

N_i molar flux of species i with respect to framework, mol m⁻² s⁻¹

p_i partial pressure of species i, Pa

q_i component molar loading of species i, mol kg⁻¹

q_0 i,mol loading of species i at saturation, mol kg⁻¹

q_i total molar loading in mixture, mol kg⁻¹

r_i(t) position vector for molecule i of species i at any time t, m

R gas constant, 8.314 J mol⁻¹ K⁻¹

T absolute temperature, K

x_i mole fraction of species i in adsorbed phase, dimensionless

z distance coordinate, m

Greek alphabet

β parameter defined by eq 13, dimensionless

δ thickness of membrane, m

δ_i Kronecker delta, dimensionless

ζ coordination number defined in eq 12, dimensionless

λ jump distance in lattice model, m

Λ[i] matrix of Maxwell–Stefan diffusivities, m² s⁻¹

µ_i molar chemical potential of component i, J mol⁻¹

π spreading pressure, N m⁻²

Π_i permeance of species i for zeolite membrane, mol m⁻² s⁻¹ Pa⁻¹

θ fractional occupancy, dimensionless

ν jump frequency, s⁻¹

ρ framework density, kg m⁻³

Subscripts

1 referring to component 1

2 referring to component 2

i referring to component i

t referring to total mixture

sat referring to saturation conditions

REFERENCES


(19) Krishna, R.; Baur, R.; Van Baten, J. M. Highlighting Diffusion Coupling Effects in Zeolite Catalyzed Reactions by Combining the


(68) Krishna, R.; van Baten, J. M. Highlighting a variety of unusual characteristics of adsorption and diffusion in microporous materials induced by clustering of guest molecules. Langmuir 2010, 26, 8450−8463.


